NANOPARTICLES

Octahedral palladium nanoparticles as excellent hosts for electrochemically adsorbed and absorbed hydrogen

Anna Zalineeva,1 Stève Baranton,1* Christophe Coutanceau,1 Gregory Jerkiewicz2*

We report new results for electrochemical H adsorption on and absorption in octahedral palladium nanoparticles (Pd-NPs) with an average tip-to-tip size of 7.8 nm and a narrow size distribution. They reveal a very high H loading of 0.90 that cannot be achieved using bulk Pd materials or larger NPs; this behavior is assigned to a combination of two factors: their small size and face morphology. Temperature-dependent cyclic voltammetry (CV) studies in the range of 296 to 333 K reveal unique features that are attributed to electrochemical H adsorption, H absorption, and H2 generation. The CV features are used to prepare H adsorption and absorption isotherms that are then used in thermodynamic data analysis. Modeling of the experimental results demonstrates that, upon H adsorption and absorption, Pd-NPs develop a core-shell-skin structure, each with its unique H loading. The electrochemical results obtained for octahedral Pd-NPs are compared to analogous data obtained for cubic Pd-NPs with a similar size as well as for larger cubic Pd-NPs and bulk materials under gas-phase conditions.

INTRODUCTION

Palladium (Pd) in the form of preferentially shaped nanoparticles (NPs) or thin layers is an excellent catalyst that finds application in a wide range of chemical reactions (1–3), and its cost makes it an attractive alternative to platinum (Pt)–based materials. Nanoparticles have a significant value of dispersion (the fraction of atoms of a material belonging to its surface), as compared to bulk materials whose dispersion values are practically zero; thus, they offer effective utilization of expensive or rare materials that have excellent catalytic properties. Pd is an excellent hydrogen (H) host and serves as a model system in research on H absorption and desorption (4). Non-noble intermetallic materials of AB3 and AB5 types are also excellent H hosts (5, 6) and find application as anodes in rechargeable nickel–metal hydride [Ni-M(H)] batteries (7). The charge and discharge kinetics of Ni-M(H) batteries are limited by the slow H diffusion in solids. This kinetic limitation can be overcome through the use of NPs that, due to their dimensions, can be quickly loaded with H. In addition, NPs offer natural resilience to pulverization and stable H storage capacity upon repetitive charge-discharge cycling. Pd is capable of both adsorbing and absorbing H, and, in the case of bulk materials, the amount of adsorbed H (Hads) is tiny as compared to the amount of absorbed H (Habs). H adsorption and absorption can be accomplished under gas-phase and electrochemical conditions, but the respective mechanisms are different owing to the nature of the H precursor (8–10). At the ambient temperature (T), H absorption under gas-phase conditions requires elevated pressures (p); the higher the pressure, the greater the amount of Habs. Under electrochemical conditions, H absorption is accomplished by applying a potential (E) at which the electrolytic H2 generation takes place (H2 generation and H absorption occur concurrently) and the amount of Habs can be related to the value of E. Pt group metals (PGMs) reveal a unique ability to adsorb H at positive overpotentials (η) with respect to the onset potential of the hydrogen evolution reaction (HER); this process is called the underpotential deposition of H (UPD H), and the species is called the underpotential deposited H (HUPD). The adsorption of H intermediate involved in the H2 generation at negative η is called the overpotential deposition of H (OPD H), and the species is called the overpotential deposited H (HOPD) (8). In the case of bulk Pd, the efficiency of H absorption in the UPD H region is 100% because it is not accompanied by any other faradaic process. On the other hand, the efficiency of H absorption in the OPD H range is lower due to the concurrent H2 generation (9). Because UPD H is accompanied by H absorption, it is impossible to determine the surface coverage (θ) of HUPD and to examine thermodynamics of the process, unlike in the case of Pt materials (8, 10). Pd nano-materials reveal higher H loading than do bulk materials (11–13), and suitable electrochemical conditions result in the separation of cyclic voltammetry (CV) features assigned to UPD H, H absorption, and HER (14).

Here, we report on the preparation of small octahedral and cubic Pd-NPs with an average size of 7.8 and 10 nm, respectively, and a narrow size distribution, followed by their application in temperature-dependent electrochemical research. An electron microscopy analysis demonstrates that these Pd-NPs maintain their shape and size after repetitive potential cycling, thus maintaining structural integrity. Then, we conduct temperature-dependent electrochemical measurements to study H adsorption on and absorption in these Pd-NPs and to analyze the energetics of these processes. The thermodynamic data are compared to analogous results obtained for bulk Pd materials to identify phenomena originating from the nanoscopic size and surface morphology of the Pd particles. This comparative analysis reveals that the octahedral Pd-NPs absorb remarkably more H than do the cubic Pd-NPs of similar size, larger NPs, or bulk Pd materials. We model the electrochemical H adsorption and absorption data to determine whether the Pd-NPs develop an inner structure that is unique to their size and observe a core-shell-skin structure. Last, we discuss that the high H loading makes the octahedral Pd-NPs very promising materials for applications such as H storage and metal hydride batteries.

RESULTS

Electrochemical H adsorption on and absorption in octahedral Pd-NPs

Figure 1A presents a high-resolution transmission electron microscopy (HR-TEM) image of octahedral Pd-NPs placed on a carbon membrane that serves as a substrate. The inset shows an image of a single NP and the corresponding fast Fourier transform (FFT) pattern. It reveals that the octahedral NPs are truncated at the extreme ends, and the FFT pattern

1Institut de Chimie des Milieux et Matériaux de Poitiers (IC2MP), UMR CNRS 7285, Equipe “Catalyse et Milieux Non Conventionnels,” Université de Poitiers, 4 rue Michel Brunet, TSA 51106, 86073 Poitiers Cedex 9, France. 2Department of Chemistry, Queen’s University, 90 Bader Lane, Kingston, Ontario K7L 3N6, Canada.
*Corresponding author. Email: steve.baranton@univ-poitiers.fr (S.B.); gregory.jerkiewicz@chem.queensu.ca (G.J.)
shows that the NPs are crystalline in nature. The main image is used to
determine the length ($l$) of the octahedral NPs, which refers to the tip-to-
tip distance. Figure 1B presents a histogram showing that the values of
$l$ fall in the range of 6 to 9 nm, with an average size of 7.8 nm, which
 corresponds to an average edge length of 5.5 nm. Figure 1C presents
identical location TEM (IL-TEM) images of the octahedral Pd-NPs
before and after potential cycling in the range of 0 to
0.40 V. The purple and red transients refer to UPD H (shown in detail in the inset), the green and blue transients refer to H absorption and $H_{abs}$ desorption, and the black transient refers to HER. RHE, reversible hydrogen electrode. (E) CV profiles for $H_{UPD}$ adsorption (shades of purple) and desorption (shades of red), and (F) CV profiles for H absorption (shades of green) and $H_{abs}$ desorption (shades of blue) for five temperature values acquired in 0.5 M aqueous $H_2SO_4$ solution at $s = 1.0$ mV s$^{-1}$.

Figure 1. Physical and electrochemical characterization of octahedral Pd-NPs. (A) HR-TEM image of octahedral Pd-NPs. The inset shows an image of a single NP and the corresponding FFT pattern. (B) Histogram showing the NP size ($l$) distribution. (C) IL-TEM images of the octahedral Pd-NPs before and after potential cycling in the range of 0 to 0.40 V. (D) CV profile for octahedral Pd-NPs acquired in 0.5 M aqueous $H_2SO_4$ solution at $T = 296$ K and $s = 1.0$ mV s$^{-1}$ in the range of $-0.05$ to 0.40 V. The purple and red transients refer to UPD H (shown in detail in the inset), the green and blue transients refer to H absorption and $H_{abs}$ desorption, and the black transient refers to HER. RHE, reversible hydrogen electrode. (E) CV profiles for $H_{UPD}$ adsorption (shades of purple) and desorption (shades of red), and (F) CV profiles for H absorption (shades of green) and $H_{abs}$ desorption (shades of blue) for five temperature values acquired in 0.5 M aqueous $H_2SO_4$ solution at $s = 1.0$ mV s$^{-1}$.

associated with these processes is small, it is shown on a different scale in the inset. The well-defined but asymmetric cathodic (green) and anodic (blue) peaks in the range of 0 to 0.12 V are due to H absorption and desorption of H_{abs}, whereas the gradually increasing current (I) at $E < -0.025$ V (black) is due to the electrolytic H$_2$ generation occurring on the surfaces of octahedral Pd-NPs now modified by H$_{UPD}$ and H$_{abs}$. The separation of CV characteristics associated with these processes is a unique feature related to the nanoscopic size of the particles because this effect is not observed in the case of bulk Pd materials (17, 18).

Thermodynamics of electrochemical H adsorption

Figure 2A presents isotherms for adsorption and desorption of H$_{UPD}$ on the octahedral Pd-NPs at temperatures in the range of 296 to 333 K acquired in 0.5 M aqueous H$_2$SO$_4$ solution at $s = 1.0$ mV s$^{-1}$. The temperature increase shifts all the features toward lower potentials and slightly modifies their shapes. The cathodic and anodic peaks are the same, and their integration (allowing for the double-layer charging) yields the charge ($Q$) values that are $Q = 400 \pm 12$ \(\mu\)C for the entire temperature range, indicating that the amount of adsorbed and desorbed H$_{UPD}$ is unaffected by T modification. Figure 1F presents a set of CV profiles for H absorption (shades of green) and H$_{abs}$ desorption (shades of blue) at the same condition, as specified above. The temperature increase shifts the cathodic and anodic peaks toward lower potentials and makes the peaks sharper. Integration of the cathodic and anodic peaks (allowing for the double-layer charging) yields the values of Q associated with H absorption and desorption of H$_{abs}$. For the entire temperature range, the value is consistently $Q = 1240 \pm 50$ \(\mu\)C, implying that the amount of H$_{abs}$ is unaffected by T modification. In addition, the amount of absorbed H equals the amount of desorbed H$_{abs}$ indicating that there is no residual H$_{abs}$ remaining in the lattice of octahedral Pd-NPs. Consequently, their charge and discharge capacities are the same. The absence of changes in the CV profiles indicates that not only the octahedral Pd-NPs maintain their size but also the NP facets do not undergo reconstruction or restructuring. For comparative analysis, we performed analogous experiments using cubic Pd-NPs with an average size of 10 nm. Figure S1 presents the following results: an HR-TEM image for cubic Pd-NPs (fig. S1A), a CV profile in the range of −0.05 to 0.40 V at $T = 293$ K (fig. S1B), a series of CV profiles for H$_{UPD}$ adsorption and desorption at four different temperature values (fig. S1C), and a series of CV profiles for H absorption and H$_{abs}$ desorption at five different temperature values (fig. S1D). The series of CV profiles were acquired in 0.5 M aqueous H$_2$SO$_4$ at $s = 1.0$ mV s$^{-1}$. The charge associated with H$_{UPD}$ adsorption and desorption is $Q = 291 \pm 11$ \(\mu\)C, and the charge due to H absorption and H$_{abs}$ desorption is $Q = 2060 \pm 84$ \(\mu\)C for the entire temperature range. As in the case of octahedral Pt-NPs, the temperature variation does not affect the amount of adsorbed or absorbed H. In a subsequent section, these CV profiles are used to prepare adsorption, absorption, and desorption isotherms that then serve in thermodynamic analyses of these processes.

Thermodynamics of electrochemical H adsorption

Figure 2A presents isotherms for adsorption and desorption of H$_{UPD}$ on the octahedral Pd-NPs at temperatures in the range of 296 to 333 K, which are prepared on the basis of the CV profiles shown in Fig. 1E. Because the H$_{UPD}$ adsorption and desorption CV profiles are not mirror images, the adsorption and desorption isotherms do not overlap, generating a hysteresis, the origin of which is discussed later. The data demonstrate that to maintain a given H$_{UPD}$ surface coverage ($\theta_{H_1}$) while increasing the temperature, the applied potential has to be decreased. Because the H$_{UPD}$ adsorption (cathodic) CV profile is asymmetric and the current (I) drops steeply as $E$ approaches the onset of H$_2$ generation, the adsorption isotherms are not S-shaped; a similar behavior is observed for the H$_{UPD}$ desorption isotherm. The application of the general electrochemical adsorption isotherm (Eq. 1) allows the determination of the standard Gibbs energy of H$_{UPD}$ electrochemical adsorption and desorption [$\Delta_{ec-ads}G^\circ(H_{UPD})$ and $\Delta_{ec-des}G^\circ(H_{UPD})$] as a function of $\theta_{H_1}$ and $T$ (19).

$$\frac{\theta_{H_{UPD}}}{1 - \theta_{H_{UPD}}} = \sqrt{f_{H_1}} \exp \left(\frac{-E_{RHE}F}{RT} \right) \exp \left(\frac{-\Delta_{ec-ads}G^\circ(H_{UPD})}{RT}\right)$$

where $f_{H_1}$ ($f_{H_1} = 1$ bar) is the fugacity of H$_2$(g) in the reference electrode compartment and $E_{RHE}$ is the potential measured with respect to RHE. Figure 2B presents graphs of $\Delta_{ec-ads}G^\circ(H_{UPD})$ and $\Delta_{ec-des}G^\circ(H_{UPD})$ as a function of $\theta_{H_1}$ for the temperatures studied. These relationships are wave-shaped and show that the values of $\Delta_{ec-ads}G^\circ(H_{UPD})$ vary between −21.3 and −14.4 kJ mol$^{-1}$ and those of $\Delta_{ec-des}G^\circ(H_{UPD})$ vary between +18.7 and +23.8 kJ mol$^{-1}$. For every $\theta_{H_1}$, the value of $\Delta_{ec-ads}G^\circ(H_{UPD})$ becomes less negative and that of $\Delta_{ec-des}G^\circ(H_{UPD})$ becomes less positive as $T$ increases. This behavior arises directly from the changes in the CV profiles brought about by $T$ increase (Fig. 1E), that is, progressively lower potentials have to be applied as temperature increases to achieve the same values of $\theta_{H_1}$ (Fig. 2A). Because the H$_{UPD}$ adsorption and desorption CV profiles are not mirror images, for a given T, the absolute values of $\Delta_{ec-ads}G^\circ(H_{UPD})$ and $\Delta_{ec-des}G^\circ(H_{UPD})$ are different. This behavior is unique to Pd-NPs because, in the case of Pt(111) and even Pt(poly) electrodes as well as Pt-NPs, the respective CV profiles are almost mirror images (8, 10, 20, 21). Because for every pair of $T$ and $\theta_{H_1}$ values the absolute value of $\Delta_{ec-des}G^\circ(H_{UPD})$ is greater than that of $\Delta_{ec-ads}G^\circ(H_{UPD})$, their sum $\delta G^\circ(H_{UPD}) = \Delta_{ec-ads}G^\circ(H_{UPD}) + \Delta_{ec-des}G^\circ(H_{UPD})$ always adopts positive values between +1.5 and +4.5 kJ mol$^{-1}$ (see fig. S2).

It is important to analyze the origin of the asymmetry in the CV profiles for H$_{UPD}$ adsorption and desorption. Because the potential scan rate is very low ($s = 1.0$ mV s$^{-1}$), we propose that the asymmetry in the CV profiles arises for nonkinetic reasons and has a thermodynamic origin. A complete CV profile refers to a close thermodynamic cycle, meaning $\delta G^\circ = 0$. The nonzero $\delta G^\circ$ might be attributed to one or more concurrently occurring interfacial processes, such as reconstruction of nanocrystalline facets ($\Delta_{reconst}G^\circ$), NP compression ($\Delta_{comp}G^\circ$), changes in interfacial hydration ($\Delta_{hydr}G^\circ$), or NP dissolution ($\Delta_{diss}G^\circ$). The reconstruction of nanocrystalline facets can be excluded (thus, $\Delta_{reconst}G^\circ = 0$) because the CV profiles do not undergo any changes for all temperatures used, and the IL-TEM images (Fig. 1C) reveal that the octahedral Pd-NPs preserve their shape and size. The IL-TEM measurements also imply that there is no dissolution of Pd-NPs; thus, $\Delta_{diss}G^\circ = 0$. This behavior is expected because the standard potentials of the Pd$^{2+}$(aq)/Pd(s) and PdO(s), H$^+(aq)/$Pd(s), H$_2$O(l) redox couples are 0.95 and 0.79 V, respectively, and the highest potential applied in this study is 0.40 V (15). Compression is not expected to play a significant role in the H$_{UPD}$ thermodynamics (thus, $\Delta_{comp}G^\circ = 0$) because the process is limited to the topmost substrate layers and does not involve the entire three-dimensional structure of Pd-NPs. Having excluded these three phenomena, we propose that the nonzero value of $\delta G^\circ(H_{UPD})$ is due to changes in the interfacial interactions of the electrolyte components (hydrated cations and H$_2$O molecules) with Pd-NPs. This proposal is supported by the observation that noncovalent interactions
between Pt materials and hydrated alkali cations \([Pt_{\text{surface}}^{-}M^{+}(aq)]\) were reported to affect kinetics of reactions occurring at Pt electrocatalysts in fuel cells by blocking active surface sites (22). Elsewhere (23, 24), it was reported that the wetting ability of Pt materials undergoes a significant change upon the adsorption of HUPD, making the surface hydrophobic—altering the strength of \(Pt_{\text{surface}}^{-}H_{2}O\) interactions. Because the \(Pt_{\text{surface}}^{-}M^{+}(aq)\) and \(Pt_{\text{surface}}^{-}H_{2}O\) interactions do not involve an external charge transfer that would give rise to a significant feature in CV transients, any change in their strength cannot be detected using this technique but can be detected indirectly through the nonzero value of \(\delta G^\circ(H_{\text{UPD}})\). The entire \(\delta G^\circ(H_{\text{UPD}})\) has two components, cathodic \([\delta G_{\text{cath}}^\circ(H_{\text{UPD}})]\) and anodic \([\delta G_{\text{anod}}^\circ(H_{\text{UPD}})]\), and \(\delta G^\circ(H_{\text{UPD}}) = \delta G_{\text{cath}}^\circ(H_{\text{UPD}}) + \delta G_{\text{anod}}^\circ(H_{\text{UPD}})\); \(\delta G^\circ(H_{\text{UPD}})\) accounts for the Gibbs energy changes associated with other phenomena (here, the interactions of the electrolyte components with Pd-NPs) occurring simultaneously with UPD H. The positive values of \(\delta G^\circ(H_{\text{UPD}})\) imply that, for all \(\theta_H\) and \(T\) values, more Gibbs energy is supplied to the system upon HUPD desorption than is released during its adsorption, as a consequence of the existence of two energetically different surface states (unmodified Pd and HUPD-modified Pd surfaces). However, because a complete CV transient commencing and ending at 0.40 V corresponds to a closed thermodynamic cycle, the sum of all individual Gibbs energy contributions is equal to zero; thus, \(\Delta_{\text{ec-ads}}G^\circ(H_{\text{UPD}}) + \Delta_{\text{ec-des}}G^\circ(H_{\text{UPD}}) + \Delta_{\text{cath}}G^\circ(H_{\text{UPD}}) + \Delta_{\text{anod}}G^\circ(H_{\text{UPD}}) = 0\).

For a given value of \(\theta_H\), the relationships between \(\Delta_{\text{ec-ads}}G^\circ(H_{\text{UPD}})\) and \(T\) or \(\Delta_{\text{ec-des}}G^\circ(H_{\text{UPD}})\) and \(T\) are linear (the correlation coefficient is 0.99), allowing the determination of the entropy of electrochemical H adsorption and desorption \([\Delta_{\text{ec-ads}}S^\circ(H_{\text{UPD}})\text{ and }\Delta_{\text{ec-des}}S^\circ(H_{\text{UPD}})]\); see fig. S3); \(\Delta_{\text{ec-ads}}S^\circ(H_{\text{UPD}})\) is negative and \(\Delta_{\text{ec-des}}S^\circ(H_{\text{UPD}})\) is positive for the entire range of \(\theta_H\). In addition, these values are more negative and more positive, respectively, than those for HUPD adsorption on Pt(111), Pt(poly), and Rh(poly) electrodes (8, 10, 19). Although both Pt and Pd adopt the face-centered cubic (fcc) structure, the more negative values of \(\Delta_{\text{ec-ads}}S^\circ(H_{\text{UPD}})\) and the more positive values of \(\Delta_{\text{ec-des}}S^\circ(H_{\text{UPD}})\) for the octahedral Pd-NPs point to a higher degree of HUPD immobilization (a stronger surface bond) in the Pd-NP lattice than in the case of bulk Pt and Rh that have practically infinite lattices. This behavior can be related to the lattice parameter of Pd-NPs that is slightly reduced (lattice contraction) as compared to bulk Pd (25). Thus, an embedded HUPD adsorbed atom resides in a slightly tighter (compressed) metallic lattice of an NP as compared to the lattice of a bulk Pd material. It is important to add that we could not compare the behavior of Pd-NPs to that of bulk Pd materials because, as we explained in the Introduction, the electrochemical H adsorption (UPD H) cannot be examined using bulk materials because of the concurrently occurring H absorption and HER. In addition, there are no thermodynamic data for UPD H on Pt-NPs and, consequently, any quantitative analysis is limited to bulk Pt and Rh materials and Pd-NPs.

Knowledge of the values of \(\Delta_{\text{ec-ads}}G^\circ(H_{\text{UPD}})\), \(\Delta_{\text{ec-des}}G^\circ(H_{\text{UPD}})\), \(\Delta_{\text{ec-ads}}S^\circ(H_{\text{UPD}})\), and \(\Delta_{\text{ec-des}}S^\circ(H_{\text{UPD}})\) allows the determination of the enthalpy of electrochemical H adsorption and desorption \([\Delta_{\text{ec-ads}}H^\circ(H_{\text{UPD}})\text{ and }\Delta_{\text{ec-des}}H^\circ(H_{\text{UPD}})]\) as a function of \(\theta_H\) for the entire range of \(T\) (Fig. 2C). The plots of \(\Delta_{\text{ec-ads}}H^\circ(H_{\text{UPD}})\) and \(\Delta_{\text{ec-des}}H^\circ(H_{\text{UPD}})\) versus \(\theta_H\) for the five \(T\) values practically overlap (they vary by less than 0.1 kJ mol\(^{-1}\)), demonstrating that, in the case of the octahedral Pd-NPs, the temperature does not affect these state functions. The values of \(\Delta_{\text{ec-ads}}H^\circ(H_{\text{UPD}})\) are between −68.2 and −33.3 kJ mol\(^{-1}\), and those of \(\Delta_{\text{ec-des}}H^\circ(H_{\text{UPD}})\) are between +44.0 and +56.3 kJ mol\(^{-1}\).

In the case of \(0.05 \leq \theta_H \leq 0.60\), \(\Delta_{\text{ec-ads}}H^\circ(H_{\text{UPD}})\) is negative, indicating that, for a given \(\theta_H\), more energy...
in the form of heat is released during H_{UPD} adsorption than absorbed during its desorption. On the other hand, in the case of 0.65 \leq \theta_{H} \leq 0.95, \delta\Delta H^\circ(H_{UPD}) is positive, indicating that, for a given \theta_{H}, less heat is released during H_{UPD} adsorption than absorbed during its desorption (fig. S4). A comparison of the values of \Delta_{ec-ads}H^\circ(H_{UPD}) and T \times \Delta_{ec-ads}S^\circ(H_{UPD}) shows that, for each T and the entire range of \theta_{H}, |\Delta_{ec-ads}H^\circ(H_{UPD})| > |T \times \Delta_{ec-ads}S^\circ(H_{UPD})|; thus, UPD H on octahedral Pd-NPs is an enthalpy-driven process.

An analogous set of results for cubic Pd-NPs, namely, H_{UPD} adsorption and desorption isotherms, plots of \Delta_{ec-ads}G^\circ(H_{UPD}), \Delta_{ec-des}G^\circ(H_{UPD}), \Delta_{ec-ads}H^\circ(H_{UPD}), and \Delta_{ec-des}H^\circ(H_{UPD}) as a function of \theta_{H} for the T values reported above are presented in fig. S5. The H_{UPD} adsorption and desorption isotherms have a slightly different shape that is attributed to the shape of NPs. The values of \Delta_{ec-ads}G^\circ(H_{UPD}) vary between −21.4 and −14.8 kJ mol⁻¹, and those of \Delta_{ec-des}G^\circ(H_{UPD}) vary between +15.7 and +22.9 kJ mol⁻¹. The values of \Delta_{ec-ads}H^\circ(H_{UPD}) are between −54.7 and −40.5 kJ mol⁻¹, and those of \Delta_{ec-des}H^\circ(H_{UPD}) are between +24.1 and +58.3 kJ mol⁻¹. A comparison of the results reveals that the magnitude of these thermodynamic state functions is similar for the two types of Pd-NPs.

Knowledge of the values of \Delta_{ec-ads}H^\circ(H_{UPD}) and \Delta_{ec-des}H^\circ(H_{UPD}) allows for the determination of the Pd−H_{UPD} surface bond energy (E_{Pd−H_{UPD}}) as a function of \theta_{H}; the values of E_{Pd−H_{UPD}} (Fig. 2D) depend only slightly on \theta_{H} and vary between +251 and +286 kJ mol⁻¹. Similar results for cubic Pd-NPs are shown in fig. S5 and demonstrate that the respective E_{Pd−H_{UPD}} values vary between +259 and +276 kJ mol⁻¹. The values of E_{Pd−H_{UPD}} for the octahedral and cubic Pd-NPs are ca. 10% higher than analogous values for bulk Pt and Rh materials, both polycrystalline and single crystals (8, 10, 19). Although, at present, we are unaware of any surface bond energy values for H_{UPD} on Pt-NPs and our discussion is limited to Pd-NPs, we propose that the increase in the strength of Pd−H_{UPD} surface bond is due to the nanoscopic size of Pd octahedrons and contraction of the Pd-NP lattice, as compared to bulk Pd materials (25). In our earlier research (8, 10, 19), we indicated that the Pt−H_{UPD} and Rh−H_{UPD} surface bond energy values matched those for chemisorbed H (H_{chem}) under gas-phase conditions and, on the basis of thermodynamic analysis, concluded that these two species are equivalent and occupy the same surface adsorption sites, although the actual adsorption mechanisms are different in electrochemical and gas-phase environments (18). The actual adsorption site of H_{UPD} remains unknown, but it is accepted that it is strongly embedded in the fcc lattice of these metals and occupies either a multifold hollow site [threefold in the case of the (111) surface and fourfold in the case of the (100) surface] in the first surface layer or an octahedral site between the two topmost surface layers. The observation that very similar bond energies are observed for H_{UPD} residing on the surfaces of octahedral and cubic Pd-NPs and bulk Pt and Rh materials leads to the conclusion that, in the case of octahedral Pd-NPs, H_{UPD} occupies the same surface adsorption site as in the case of bulk materials. Because the Pd-NPs are octahedral and have predominantly (111) facets, we propose that H_{UPD} occupies either a threefold hollow site in the first surface layer or an octahedral site between the two topmost surface layers.

In the case of perfect octahedral Pd-NPs, all facets have the (111) orientation. Our TEM analysis indicates that the Pd-NPs are not perfect and lack two to three atomic layers at the corners and one to two atomic layers along the edges. Such modified corners and edges mimic the (100) and (110) structures, respectively. However, because they account for a tiny fraction of the overall surface area, their contribution to the overall electrochemical signals is negligible. Above, we determine thermodynamic state functions for the electrochemical H adsorption on Pd-NPs and compare them to analogous data for bulk Pt and Pt(111) electrodes obtained also on the basis of temperature-dependent studies. At this stage of the discussion, it is important to add that there are no equivalent results for the Pt(100) or Pt(110) electrodes. Consequently, it is impossible to compare the thermodynamic data presented above to analogous results obtained for other Pt monocrystalline electrodes than Pt(111) or Pt(poly).

**Thermodynamics of electrochemical H absorption**

The temperature-dependent CV profiles (Fig. 1F) reveal well-defined features for H adsorption and H_{abs} desorption. Because they do not overlap those assigned to UPD H or HER, they create a basis for the determination of H absorption and H_{abs} desorption isotherms as well as their thermodynamic analysis. Figure 3 (A and B) presents plots of H absorption and H_{abs} desorption isotherms expressed as E versus X_{H} and f_{H} versus X_{H}, where X_{H} is the lattice occupancy fraction defined as X_{H} = N_{H}/N_{Pd,lim}N_{H} and N_{Pd,lim} are the numbers of Habs and inner Pd atoms per octahedral Pd-NP, respectively (see the Supplementary Materials). Because the surface Pd atoms participate in UPD H, only the inner atoms are involved in H absorption. The conversion of E values at which a given X_{H} is achieved to equivalent f_{H} values relates our findings to those for H absorption and H_{abs} desorption under gas-phase conditions (11). This conversion uses the Nernst equation and takes into account the mean activity coefficient of hydrated proton and other parameters (see the Supplementary Materials). The maximum H loading in octahedral Pd-NPs is found to be temperature-independent and corresponds to X_{H} = 0.90. The isotherms reveal a broad but sloped plateau as in the case of bulk materials and a hysteresis (4); the hysterisis implies that for given X_{H} and T, a higher value of f_{H} is required to drive H absorption than H_{abs} desorption. The plateau corresponds to the coexistence of the α and β phases and represents the transition from the α phase to the β phase during H absorption and from the β phase to the α phase during H_{abs} desorption. Elsewhere (26), it is reported that, in the case of single Pd-NPs, the plateau is practically horizontal, whereas it is sloped in the case of an ensemble of NPs. In our case, the sloped plateau is expected because we report results for an ensemble of Pd-NPs with a certain size distribution (see Fig. 1B). The E versus X_{H} and f_{H} versus X_{H} isotherms become practically vertical when X_{H} reaches 0.90, indicating that X_{H} = 0.90 corresponds to the maximum loading of H_{abs} and the application of even lower potentials (thus, higher equivalent f_{H} values) does not increase it any further. In fig. S6 (A and B), the graphs present E versus X_{H} and f_{H} versus X_{H} absorption and desorption isotherms prepared on the basis of the results shown in fig. S1. The results reveal that, in the case of cubic Pd-NPs, the maximum H loading is X_{H} = 0.66, thus substantially lower than in the case of the octahedral NPs. The isotherms reveal a broad but sloped plateau corresponding to the coexistence of the α and β phases. However, in the case of cubic Pd-NPs, significantly higher values of f_{H} are required to accomplish the same value of X_{H} (but still lower than 0.66), as in the case of octahedral Pd-NPs.

At this stage of the discussion, it is important to discuss the relationship between the potential of the H electrode and the H_{2} fugacity (the effective H_{2} pressure) as they appear in the Nernst equation for a specific activity of the hydrated H^{+}. The standard potential refers to the H_{2} fugacity being equal to the standard pressure (p^p = 1 bar) and the activity of H^{+} being one. The Nernst equation represents an equilibrium between the H_{2} fugacity above the electrolyte solution and the potential experienced by an electrode immersed in it. Positive potentials with
respect to the standard potential of the H+/H2 redox couple \( E^o(H^+/H_2) \) imply an external H2 fugacity lower than \( p^o \), and negative potentials with respect to \( E^o(H^+/H_2) \) imply an external H2 fugacity higher than \( p^o \). The H2 fugacity refers to the effective pressure of H2 above the electrolyte solution (27).

In situ TEM–electron energy loss spectroscopy was used by others to study Pd hydride formation and revealed that, in the case of single cubic Pd-NPs that have side lengths in the range of 13 to 65 nm, the \( \alpha \)-to-\( \beta \) phase transition plateau at \( T = 246 \) K corresponds to H2(g) pressure that is in the range of 10 to 100 Pa (11). Although the actual H loading was not measured, the authors performed calculations on the assumption that \( X_{H1} \) was in the range of 0.60 to 0.70. Our new results demonstrate that, under electrochemical conditions, the octahedral Pd-NPs can absorb significant amounts of H and values as high as \( X_{H1} = 0.90 \) can be achieved. In addition, the maximum loading of \( X_{H1} = 0.90 \) can be reached at lower equivalent pressures than those reported in the literature (11, 28, 29). We propose that this unique behavior can be assigned to the octahedral shape of Pd particles and their nanoscopic size. Specifically, (111) facets dominate the entire structure of the octahedral Pd-NPs, although the edges mimic (110) facets but make a small contribution to the overall surface area. Because the surface coordination numbers of fcc(111) and fcc(100) are 9 and 8, respectively, the surface tension of the fcc(111) face is smaller than that of the fcc(100) face. Absorption of H gives rise to an expansion of the Pd lattice, which is opposed by the surface tension. At this stage of our analysis, we propose that, in the case of small octahedral Pd-NPs, the counteracting lattice expansion due to H absorption and lattice compression due to the surface tension create a structure that favors significantly higher H loading than in the case of cubic NPs or bulk Pd materials. The large increase (30 to 50%) in the H loading at lower equivalent H2(g) pressures as compared to other Pd nanomaterials and bulk materials makes octahedral Pd-NPs very promising materials for possible future applications such as miniaturized H storage devices and metal hydride batteries.

Figure 3C presents \( \Delta_{H}^f(H_{abs}) \) versus 1/T plots for 0.10 \( \leq X_{H1} \leq 0.80 \), with an interval of \( \Delta X_{H1} = 0.10 \) (calculations were performed for an interval of \( X_{H1} = 0.05 \), but for clarity of presentation, these additional plots are not shown). These relationships are linear (the correlation coefficient is at least 0.98), and their slopes, which are determined through the application of Eq. 2, make the determination of the enthalpy of electrochemical H absorption and H abs desorption \( [\Delta_{ec-abs}H^f(H_{abs})] \) and \( \Delta_{ec-des}H^f(H_{abs}) \) possible.

\[
\frac{\partial (\ln \sqrt{f_{H2}/p^o})}{\partial T} \bigg|_{X_{H1}} = -\frac{\Delta_{ec-abs}H^f(H_{abs})}{R \ T^2}
\]

and

\[
\frac{\partial (\ln \sqrt{f_{H2}/p^o})}{\partial T} \bigg|_{X_{H1}} = \frac{\Delta_{ec-des}H^f(H_{abs})}{R \ T^2}
\]

where \( p^o \) is the standard pressure and \( R \) is the ideal gas constant. Figure 3D shows plots of \( \Delta_{ec-abs}F^f(H_{abs}) \) and \( \Delta_{ec-des}F^f(H_{abs}) \) as a function of \( X_{H1} \) and demonstrates that \( \Delta_{ec-abs}F^f(H_{abs}) \) adopts values between \(-19.6 \) and \(-29.4 \) kJ mol\(^{-1}\), whereas \( \Delta_{ec-des}F^f(H_{abs}) \) adopts values between \(+15.9 \) and \(+46.3 \) kJ mol\(^{-1}\). Because for a given \( X_{H1} \) the absolute values of \( \Delta_{ec-abs}F^f(H_{abs}) \) and \( \Delta_{ec-des}F^f(H_{abs}) \) are different, their sum, defined as \( \delta\Delta F^f(H_{abs}) = \Delta_{ec-abs}F^f(H_{abs}) + \Delta_{ec-des}F^f(H_{abs}) \), is nonzero and adopts the standard potential of the H+/H2 redox couple \( E^o(H^+/H_2) \) imply an external H2 fugacity lower than \( p^o \), and negative potentials with respect to \( E^o(H^+/H_2) \) imply an external H2 fugacity higher than \( p^o \). The H2 fugacity refers to the effective pressure of H2 above the electrolyte solution (27).

In situ TEM–electron energy loss spectroscopy was used by others to study Pd hydride formation and revealed that, in the case of single cubic Pd-NPs that have side lengths in the range of 13 to 65 nm, the \( \alpha \)-to-\( \beta \) phase transition plateau at \( T = 246 \) K corresponds to H2(g) pressure that is in the range of 10 to 100 Pa (11). Although the actual H loading was not measured, the authors performed calculations on the assumption that \( X_{H1} \) was in the range of 0.60 to 0.70. Our new results demonstrate that, under electrochemical conditions, the octahedral Pd-NPs can absorb significant amounts of H and values as high as \( X_{H1} = 0.90 \) can be achieved. In addition, the maximum loading of \( X_{H1} = 0.90 \) can be reached at lower equivalent pressures than those reported in the literature (11, 28, 29). We propose that this unique behavior can be assigned to the octahedral shape of Pd particles and their nanoscopic size. Specifically, (111) facets dominate the entire structure of the octahedral Pd-NPs, although the edges mimic (110) facets but make a small contribution to the overall surface area. Because the surface coordination numbers of fcc(111) and fcc(100) are 9 and 8, respectively, the surface tension of the fcc(111) face is smaller than that of the fcc(100) face. Absorption of H gives rise to an expansion of the Pd lattice, which is opposed by the surface tension. At this stage of our analysis, we propose that, in the case of small octahedral Pd-NPs, the counteracting lattice expansion due to H absorption and lattice compression due to the surface tension create a structure that favors significantly higher H loading than in the case of cubic NPs or bulk Pd materials. The large increase (30 to 50%) in the H loading at lower equivalent H2(g) pressures as compared to other Pd nanomaterials and bulk materials makes octahedral Pd-NPs very promising materials for possible future applications such as miniaturized H storage devices and metal hydride batteries.

Figure 3C presents \( \ln f_{H1} \) versus 1/T plots for 0.10 \( \leq X_{H1} \leq 0.80 \), with an interval of \( \Delta X_{H1} = 0.10 \) (calculations were performed for an interval of \( X_{H1} = 0.05 \), but for clarity of presentation, these additional plots are not shown). These relationships are linear (the correlation coefficient is at least 0.98), and their slopes, which are determined through the application of Eq. 2, make the determination of the enthalpy of electrochemical H absorption and H abs desorption \( [\Delta_{ec-abs}H^f(H_{abs})] \) and \( \Delta_{ec-des}H^f(H_{abs}) \) possible.

\[
\frac{\partial (\ln \sqrt{f_{H2}/p^o})}{\partial T} \bigg|_{X_{H1}} = -\frac{\Delta_{ec-abs}H^f(H_{abs})}{R \ T^2}
\]

and

\[
\frac{\partial (\ln \sqrt{f_{H2}/p^o})}{\partial T} \bigg|_{X_{H1}} = \frac{\Delta_{ec-des}H^f(H_{abs})}{R \ T^2}
\]

where \( p^o \) is the standard pressure and \( R \) is the ideal gas constant. Figure 3D shows plots of \( \Delta_{ec-abs}F^f(H_{abs}) \) and \( \Delta_{ec-des}F^f(H_{abs}) \) as a function of \( X_{H1} \) and demonstrates that \( \Delta_{ec-abs}F^f(H_{abs}) \) adopts values between \(-19.6 \) and \(-29.4 \) kJ mol\(^{-1}\), whereas \( \Delta_{ec-des}F^f(H_{abs}) \) adopts values between \(+15.9 \) and \(+46.3 \) kJ mol\(^{-1}\). Because for a given \( X_{H1} \) the absolute values of \( \Delta_{ec-abs}F^f(H_{abs}) \) and \( \Delta_{ec-des}F^f(H_{abs}) \) are different, their sum, defined as \( \delta\Delta F^f(H_{abs}) = \Delta_{ec-abs}F^f(H_{abs}) + \Delta_{ec-des}F^f(H_{abs}) \), is nonzero and adopts...
mainly positive values that gradually decrease from the highest value of +18.3 kJ mol$^{-1}$ to the only negative value of −3.7 kJ mol$^{-1}$ for $X_{H_{1}} = 0.85$ (fig. S7). The main positive values of $\Delta H_{f}(H_{ab})$ indicate that the heat absorbed during $H_{ab}$ desorption is greater than the heat released during $H$ absorption. In fig. S6C, the graph presents $f_{H_{1}}$ versus $1/T$ for $0.10 \leq X_{H_{1}} \leq 0.65$, with an interval of $\Delta X_{H_{1}} = 0.05$, and in fig. 6D, the graph plots $\Delta_{ec-ab}H^f(H_{ab})$ and $\Delta_{dec-des}H^f(H_{ab})$ as a function of $X_{H_{1}}$ for cubic Pd-NPs. They demonstrate that $\Delta_{ec-ab}H^f(H_{ab})$ adapts values between −12.0 and −5.3 kJ mol$^{-1}$, whereas $\Delta_{dec-des}H^f(H_{ab})$ adopts values between +9.5 and +28.4 kJ mol$^{-1}$. Although the Pd-H system has been extensively investigated, most of the thermodynamic data refer to bulk materials, and there are few studies dedicated to $H$ absorption in Pd nanomaterials. In situ luminescence probe studies of $H$ absorption in cubic Pd-NPs that have side lengths in the range of 14 to 110 nm resulted in the determination of $\Delta_{ec-ab}H^f(H_{ab})$ that varies from −13.7 kJ mol$^{-1}$ for the smallest NPs to −16.4 kJ mol$^{-1}$ for the largest ones (the original data that report enthalpy values per mole of $H_{2}$ are converted to enthalpy values per 1 mole of $H_{ab}$ (29). Analogous data for $H$ absorption and $H_{ab}$ desorption in bulk Pd materials are $\Delta_{ec-ab}H^f(H_{ab}) = −18.2$ kJ mol$^{-1}$ and $\Delta_{dec-des}H^f(H_{ab}) = +20.6$ kJ mol$^{-1}$ (30). Our results indicate that, due to the nanoscopic size of the Pd particles and their octahedral shape, $H$ absorption is more exothermic and $H_{ab}$ desorption is more endothermic than in the case of bulk Pd materials or cubic Pd-NPs. Knowledge of the heat evolved during $H$ absorption and $H_{ab}$ desorption is of importance to nanotechnology, where thermal requirements (for example, heat capacity) are needed to take into account in the design of nanoscopic devices. The enthalpy values reported here are very accurate because (i) electrochemical methods offer very precise determination of the amount of adsorbed $H$ ($H_{UPD}$) and absorbed $H$ ($H_{abs}$) by integrating electrochemical methods offer very precise determination of the cathodic and anodic CV profiles, (ii) applied potential values can be combined with thermodynamic equations facilitate the determination of enthalpy of $H$ absorption and $H_{abs}$ desorption for a broad range of $X_{H_{1}}$ values.

The entropy of $H$ absorption and $H_{ab}$ desorption $[\Delta_{ec-ab}S^o(H_{ab})$ and $\Delta_{dec-des}S^o(H_{ab})]$ can be determined by applying the van’t Hoff equation (Eq. 2) to the results presented in Fig. 3C. The values of $\Delta_{ec-ab}S^o(H_{ab})$ are negative and increase almost linearly from −58.7 to −33.9 J mol$^{-1}$ K$^{-1}$ with increasing $X_{H_{1}}$. The values of $\Delta_{dec-des}S^o(H_{ab})$ decrease non-linearly from +91.9 to +19.5 J mol$^{-1}$ K$^{-1}$ with increasing $X_{H_{1}}$ (fig. S8). Knowledge of the entropy values makes possible the determination of the Gibbs energy of $H$ absorption and $H_{ab}$ desorption $[\Delta_{ec-ab}G^o(H_{ab})$ and $\Delta_{dec-des}G^o(H_{ab})]$, as well as their sum defined as $\Delta G^o(H_{ab}) = \Delta_{ec-ab}G^o(H_{ab}) + \Delta_{dec-des}G^o(H_{ab})$ as a function of $X_{H_{1}}$ (fig. S9). For the five temperatures, the values of $\Delta_{ec-ab}G^o(H_{ab})$ are consistently negative and between −12.0 and −8.3 kJ mol$^{-1}$, whereas the values of $\Delta_{dec-des}G^o(H_{ab})$ are positive and between +14.9 and +9.4 kJ mol$^{-1}$. For each temperature, the values of $\Delta_{ec-ab}G^o(H_{ab})$ increase with increasing $X_{H_{1}}$, and those of $\Delta_{dec-des}G^o(H_{ab})$ decrease with increasing $X_{H_{1}}$, indicating that the averaged interactions between $H_{ab}$ atoms are repulsive. In the case of $0.10 \leq X_{H_{1}} \leq 0.80$, the $\Delta_{ec-ab}G^o(H_{ab})$ versus $X_{H_{1}}$ and $\Delta_{dec-des}G^o(H_{ab})$ versus $X_{H_{1}}$ plots are linear, pointing to a Frumkin-like behavior. Finally, the values of $\Delta G^o(H_{ab})$, which are a measure of the absorption-desorption hysteresis, are positive and small for the entire range of $X_{H_{1}}$ and all five temperatures. In an important contribution (26), it was proposed that the hysteresis in the $f_{H_{1}}$ versus $X_{H_{1}}$ plots, which is even observed in the case of single Pd-NPs, arises on the basis of an energetic interplay associated with the formation of dislocations and the coherency strain that develops at the metal/metal hydride interface during the hydride forma-

**Mechanism of $H$ absorption in octahedral Pd-NPs**

In Fig. 1D, the CV profile points to a unique behavior of Pd-NPs in the sense that UPD $H$, $H$ absorption, and HER occur in distinct potential ranges. Because $\Delta_{ec-ab}G^o(H_{UPD})$ is more negative than $\Delta_{ec-ab}G^o(H_{abs})$, $H_{UPD}$ does not undergo transition to become $H_{abs}$ as in the case of bulk Pd materials (9). Elsewhere (10, 18, 19), it was proposed that in the case of Pt(111) or Rh(111) electrodes, $H_{UPD}$ occupies either the octahedral site (Oh) between the first and the second surface monolayer (ML) or the threefold hollow site right above the Oh site but while still being embedded in the surface lattice; $H_{UPD}$ in this site is referred to as fcc(111)-H$_{UPD}(Oh)$ (fig. S9). If a complete ML of $H_{UPD}$ atoms occupies all the Oh sites, then $H$ absorption can proceed only through the adjacent tetrahedral sites (Td) between the first and second ML of Pd atoms. The adsorbed $H$ atom in the Td site referred to as fcc(111)-H$_{abs}(Td)$ is a short-lived intermediate state due to lateral repulsions; it undergoes transition to become $H_{abs}$ and eventually occupies the vacant interstitial sites beneath the second ML of Pd atoms.

Gas-phase $H$ absorption in Pd materials can be modeled using the surface stress model described elsewhere (30) and can be adapted to Pd-NPs (11). It is based on an assumption that, upon $H$ absorption, Pd-NPs develop a core-shell structure, with each component having its unique $H$ loading. The $H$ intake in the shell is fast, and this region quickly reaches its maximum $H$ loading and only then that $H$ becomes absorbed in the core. The model leads to Eqs. 3 and 4 that relate the overall $H$ loading ($X_{H_{1}}$) and $H$ loading in the shell ($X_{H_{shell}}$) to $f_{H_{1}}$, $T$, the NP diameter, and the shell thickness.

\[
\frac{RT}{2} \ln \left( \frac{f_{H_{1}}}{P} \right) = \Delta H^o_{H_{1}} - T \Delta S^o_{H_{1}} + RT \ln \left( \frac{X_{H_{1}}}{1 - X_{H_{1}}} \right) + \Delta \mu_{H_{1}} - \Delta \mu_{H_{1}} + \Delta \mu_{H_{1}} - \Delta \mu_{H_{1}}f_{H_{1}} \tag{3}
\]

\[
\Delta \mu_{H_{1}} = \frac{-3K_{H_{1}}^{-1}(X_{H_{shell}} - X_{H_{1}})}{3V_{Pd} + V_{H}X_{H_{1}}} \tag{4}
\]

\[
\Delta \mu_{H_{1}} = \frac{2d}{3(1 + d)^{2}} \tag{4}
\]

The variables appearing in these equations and the values of physical parameters required to perform simulations are provided in the Supplementary Materials. The application of this model to our experimental data yields a shell thickness of $t = 0.817$ nm and a maximum $H$ loading in the shell ($X_{H_{shell}} = 1$). The shell thickness of 0.817 nm corresponds to the three atomic layers of Pd. The model does not distinguish between $H_{UPD}$ and $H_{abs}$, because both are in the Pd lattice, and it treats them as $H$ atoms occupying interstitial sites. Bearing in mind the proposal that $H_{UPD}$ species occupy the octahedral sites beneath the first Pd monolayer (Fig. 4A), the subsequent two layers of $H_{abs}$ occupy the interstitial sites beneath the second and third Pd monolayers (Fig. 4B). The experimentally determined overall maximum $H$ loading of $X_{H_{1}} = 0.90$ and the
shell loading of $X_{\text{H,shell}} = 1$ together imply that the core loading equals $X_{\text{H,core}} = 0.86$ (Fig. 4C). A schematic representation of a single octahedral Pd-NP that has reached a maximum H loading of $X_{\text{H}} = 0.90$ and has a core-shell structure is presented as a cross section in Fig. 5A. The surface stress model can be used to calculate a set of $f_{\text{H2}}$ and $T$ values required for Pd-NPs to reach the maximum H loading of $X_{\text{H}} = 0.90$. In Fig. 5B, the solid black line presents the calculated values of $\ln \left( \frac{f_{\text{H2}}}{p_0} \right)$ as a function of $T$, whereas the red points refer to our data. The agreement indicates that the surface stress model can be successfully used to model H absorption in small Pd-NPs under electrochemical conditions.

Fig. 4. Visual representation of the different steps of HUPD adsorption and H absorption in octahedral Pd-NPs. (A) HUPD species that occupy the octahedral sites beneath the first Pd surface layer. (B) $H_{\text{abs}}$ beneath the second and third Pd monolayers; HUPD, $H_{\text{absy}}$ and the four topmost Pd layers that together form the shell region. (C) $H_{\text{abs}}$ in the core of the Pd-NP.

Fig. 5. Surface stress model for H absorption in and $H_{\text{abs}}$ desorption from octahedral Pd-NPs. (A) Visual representation of the cross section of a single octahedral Pd-NP loaded with H to $X_{\text{H}} = 0.90$ showing HUPD beneath the first Pd layer, $H_{\text{abs}}$ in the shell region, and $H_{\text{abs}}$ in the NP core. (B) Comparison of the calculated $H_{\text{f}}(g)$ fugacity values (black line) to the experimentally determined data (red points) required to reach $X_{\text{H}} = 0.90$. 

$E = 0.10 \, \text{V}$  
$H_{\text{UPD}}, \theta_{\text{H}} = 1$

$0.02 \, \text{V} < E < 0.10 \, \text{V}$  
$H_{\text{UPD}}, \theta_{\text{H}} = 1$;  
$H_{\text{abs}}, X_{\text{H,shell}} = 1; X_{\text{H,core}} = 0$

$0.00 \, \text{V} \leq E \leq 0.02 \, \text{V}$  
$H_{\text{UPD}}, \theta_{\text{H}} = 1$;  
$H_{\text{abs}}, X_{\text{H,shell}} = 1; X_{\text{H,core}} = 0.86$
DISCUSSION
In summary, small octahedral Pd-NPs that have an average size of 7.8 nm can be used as H host materials. Under electrochemical conditions at room or elevated temperatures, they can be repetitively charged with H and discharged without any modification to their shape or size. Because of their nanoscopic size, the charging and discharging are quickly achieved, and no residual absorbed H remains in the Pd nanolattice. Pulverization of bulk H-storing materials is an important technological challenge that limits the life cycle of Ni-M(H) batteries. The lack of any structural changes in the octahedral Pd-NPs upon H absorption and H desorption suggests that the degradation (pulverization) of bulk materials is due to the presence of grain boundaries and other structural defects. The octahedral Pd-NPs give rise to a new behavior: the separation of voltammetry features associated with H absorption, H desorption, and H2 generation. Together, this property and temperature-dependent experimental research make the analysis of thermodynamic and kinetic parameters of the three processes possible. Electrochemical measurements offer precise control of the applied potential and exact determination of the amount of absorbed and absorbed H. Consequently, the analysis of electrochemical H absorption and absorption yields accurate values of thermodynamic state functions. The nanoscopic nature of the Pd particles (a reduced lattice parameter) results in a higher degree of immobilization of electrochemically adsorbed H and a stronger surface bond as compared to bulk H-absorbing PGMs. Because of the nanoscopic nature of the Pd particles and their octahedral shape that gives rise to predominantly (111) surface orientations of atoms, the absorption of H is more exothermic than in the case of bulk Pd materials or cubic Pd-NPs of similar size. It is an important new piece of information because the performance and lifetime of miniaturized energy-storing devices are related to their heat capacity, and excessive heat evolution can lead to their gradual failure. Although this contribution deals only with octahedral and cubic Pd-NPs of similar size, it is conceivable that a similar analysis could be performed for Pd-NPs of different shapes and dimensions. A systematic experimental approach could result in the identification of a critical dimension and a preferred shape, which, together, give rise to size- and structure-dependent phenomena. The mechanism of electrochemical H absorption in Pd-NPs differs from that observed in the case of bulk materials because the adsorbed H (UPD H) does not undergo transition to become adsorbed H. In addition, the external shell of the octahedral Pd-NPs becomes saturated with H, and it is only then that the core starts absorbing H. Upon H absorption, the Pd-NPs develop a unique core-shell-skull structure, where the shell-skull has a maximum H loading of XSH-skull = 1.00 and the core has an H loading of XH-core = 0.86. The overall maximum H loading, which is XH = 0.90, is assigned to the Pd-NP shape and size. This H loading exceeds, by ca. 30 to 50%, the H loading capacity of bulk Pd materials or similar and larger cubic Pd-NPs. The structural integrity of the octahedral Pd-NPs and their exceptionally high H loading capacity make them very promising materials for applications such as miniaturized H storage devices and metal hydride batteries.

MATERIALS AND METHODS
Synthesis of the octahedral Pd-NPs
Octahedral Pd-NPs were synthesized using a method described elsewhere (14). This method was based on chemical reduction of K2PdCl4 (17.6 mM) in ultrahigh-purity water using polyvinylpyrrolidone (86 mM) as a surfactant and a mixture of ascorbic acid (85 mM) and citric acid (85 mM) acting as reducing and surface agents.

IL-TEM measurements
IL-TEM measurements were performed using an ultrahigh-resolution JEOL JEM-2100 microscope with a resolution of 0.19 nm. Pd-NPs were placed on a 300-mesh gold grid with a marker for identical position finding. After an IL-TEM image was acquired, the gold grid covered with octahedral Pd-NPs was used as a working electrode in electrochemical experiments. Typically, 10 CV profiles were recorded in the range of 0 ≤ E ≤ 0.40 V to observe features for electrochemical H adsorption and absorption (see below). After electrochemical measurements, the gold grid covered with octahedral Pd-NPs was rinsed with ultrahigh-purity water and was transferred to the microscope for post-electrochemical IL-TEM measurements.

Electrochemical measurements
CV experiments were performed in 0.5 M aqueous H2SO4 solution out-gassed by bubbling ultrahigh-purity N2(g). They were conducted at a potential scan rate of s = 1.0 mV s−1 and at different temperatures in the range of 296 ≤ T ≤ 333 K. The temperature was controlled using a Haake water bath; the temperature readings inside the cell and the bath agreed to ±0.5 K. The working electrode was a polycrystalline Au disc polished to a mirror-like finish on which 14 g of unsupported octahedral Pd-NPs was deposited. A glassy carbon plate (surface area of ca. 4 cm2) was used as a counter electrode. A Pt/Pt black RHE placed in a separate compartment served as a reference electrode. It was connected to the main cell compartment via a Luggin capillary. All potential values were measured and are reported with respect to RHE.

SUPPLEMENTARY MATERIALS
Supplementary material for this article is available at http://advances.sciencemag.org/cgi/content/full/3/2/e1600542/DC1
Results
fig. S1. TEM image for cubic Pd-NPs and CV profiles of HUPD adsorption and desorption, H absorption, and H desorption.
fig. S2. Plots of δG°(HUPD) as a function of T for the five temperature values.
fig. S3. Plots of δG°(HUPD)(purple) and δG°(HUPD)(red) as a function of T.
fig. S4. Plots of δG°(HUPD) as a function of T for the five temperature values.
fig. S5. Adsorption and desorption isotherms for HUPD; plots of H°(Hupd) (red) as a function of XH for the five temperature values.
fig. S6. H absorption and H desorption isotherms, van’t Hoff plots, and plots of δG°(HUPD) and δG°(HUPD) as a function of XH for the five temperature values.
fig. S7. Plot of δG°(HUPD) as a function of XH.
fig. S8. Plots of δG°(HUPD) and δG°(HUPD) as a function of XH.
fig. S9. Plots of δG°(HUPD) and δG°(HUPD) as a function of XH.
fig. S10. Visualization of the Oh and Td sites on the fcc(111) surface.
fig. S11. Variation of H°(Hupd) (red) as a function of XH.
table S1. Properties of 0.50 M aqueous H2SO4 solution.

REFERENCES AND NOTES
Hydrogen-induced phase transitions in individual palladium nanocrystals.


Hydrogen in Metals I (Springer-Verlag, 1978).


In situ detection of hydrogen-induced phase transitions in individual palladium nanocrystals.


In situ detection of hydrogen-induced phase transitions in individual palladium nanocrystals.


Hydrogen in Metals II (Springer-Verlag, 1978).


Acknowledgments: We thank S. Pronier for performing IL-TEM measurements at the Université de Poitiers. Funding: A.Z. acknowledges financial support toward her postdoctoral studies from the County Council of Poitou-Charentes, France. G.J. acknowledges support from the Natural Sciences and Engineering Research Council of Canada. Author contributions: S.B., C.C., and G.J. conceived the idea. A.Z. and S.B. synthesized the materials and conducted experiments. S.B., C.C., and G.J. performed the thermodynamic analysis and co-wrote the manuscript. Competing interests: The authors declare that they have no competing interests. Data and materials availability: All data needed to evaluate the conclusions in the paper are present in the paper and/or the Supplementary Materials. Additional data related to this paper may be requested from the authors.

Submitted 14 March 2016
Accepted 19 December 2016
Published 3 February 2017
10.1126/sciadv.1600542


Octahedral palladium nanoparticles as excellent hosts for electrochemically adsorbed and absorbed hydrogen
Anna Zalineeva, Stève Baranton, Christophe Coutanceau and Gregory Jerkiewicz

Sci Adv 3 (2), e1600542.
DOI: 10.1126/sciadv.1600542

ARTICLE TOOLS http://advances.sciencemag.org/content/3/2/e1600542
SUPPLEMENTARY MATERIALS http://advances.sciencemag.org/content/suppl/2017/01/30/3.2.e1600542.DC1
REFERENCES This article cites 25 articles, 1 of which you can access for free http://advances.sciencemag.org/content/3/2/e1600542#BIBL
PERMISSIONS http://www.sciencemag.org/help/reprints-and-permissions

Use of this article is subject to the Terms of Service

Science Advances (ISSN 2375-2548) is published by the American Association for the Advancement of Science, 1200 New York Avenue NW, Washington, DC 20005. 2017 © The Authors, some rights reserved; exclusive licensee American Association for the Advancement of Science. No claim to original U.S. Government Works. The title Science Advances is a registered trademark of AAAS.