Resonant laser printing of structural colors on high-index dielectric metasurfaces

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Man-made structural colors, which originate from resonant interactions between visible light and manufactured nanostructures, are emerging as a solution for ink-free color printing. We show that non-iridescent structural colors can be conveniently produced by nanostructures made from high-index dielectric materials. Compared to plasmonic analogs, color surfaces with high-index dielectrics, such as germanium (Ge), have a lower reflectance, yielding a superior color contrast. Taking advantage of band-to-band absorption in Ge, we laser-postprocess Ge color metasurfaces with morphology-dependent resonances. Strong on-resonance energy absorption under pulsed laser irradiation locally elevates the lattice temperature (exceeding 1200 K) in an ultrashort time scale (1 ns). This forms the basis for resonant laser printing, where rapid melting allows for surface energy changes with associated modification of color appearance. Laser-printable high-index dielectric color metasurfaces are scalable to a large area and open a new paradigm for printing and decoration with nonfading and vibrant colors.

INTRODUCTION

Optical metasurfaces (1, 2) provide new means for manipulation of optical fields with the aid of structural resonances and/or artificial birefringence that can alter amplitude, phase, and polarization over subwavelength distances (3–5). Metasurfaces are harvested from subwavelength structures that are metallic (6–9), dielectric (10–18), or hybrid combinations (19–22). Color metasurfaces generated from resonant interactions between visible light and the nanoscale surface textures are highly desired for ink-free color decoration. Thin-film interference at metasurfaces with absorbing materials has been used for color patterning (23–25), although it is inherently limited in its spatial resolution. In contrast, plasmonic color metasurfaces (26–28) have been printed with subwavelength resolution (29–32), whereas structural colors caused by the composition of low-index dielectric materials require long-range orders (33, 34), thus posing a challenge for high-resolution, large-area, and non-iridescent applications of all-dielectric color metasurfaces.

All-dielectric structures form the basis for a rich palette of structural color phenomena in nature (35, 36), and artificial colors can also be supported by carefully engineered and nanofabricated high-index dielectric surfaces (37, 38), for example, localized resonances in dielectrics (39), such as silicon (Si) nanostructures (40–43). At the same time, there is also a tremendous attention to plasmonic structural colors (28, 44), where colors develop when ultrathin metal films are evaporated onto pre-nanostructured polymer substrates, such as large arrays of polymer nanopillars (29, 32, 45). Here, we describe how essentially similar mass production approaches (46) can also be used for all-dielectric color metasurfaces by substituting metal coatings with high-index dielectric films, such as semiconductor materials.

RESULTS

A schematic representation of the structures is provided in Fig. 1A, where a thin film of germanium (Ge) is deposited on a substrate with nanoimprinted polymer pillars. In this way, dielectric disks can be raised above a same-thickness dielectric film representing an inverse of the disk array, that is, a high-index dielectric film with “holes” penetrated by low-index pillars. We note that both disks and holes inherit their common topology from the polymer pillars, and thus, one can also make other cross sections, such as bars. Ge was selected in the initial applications because of its high refractive index and because it comprises some attenuation at visible frequencies, which is essential for the ultimate color generation and laser printing that we describe. In particular, abundant high-index materials can be used for these structural color generations, which are multifunctional and potentially environment-inert (see also the results with Si in fig. S1) (47–51). Here, polymer substrates were preferred because of their flexibility, ease of fabrication, and throughput scale-up in nanoimprint lithography. Optically transparent polymers (here, Ormocomp is considered) can further enhance the color visibility and, at the same time, isolate the laser-induced heating within the on-top lossy materials in the following laser printing processes. The scanning electron microscope (SEM) image of a small selected area of the nanostructures after Ge deposition is shown in Fig. 1B, illustrating the fixed periodicity \( P = 200 \) nm of the lattice and the radius \( R \) of the disks in our design. The interaction of light within the hybrid disk-hole structure can be enhanced markedly by trapping light with multiple internal reflections. Dipole resonance conditions can be fulfilled for subwavelength structures that are made of high-index materials (40, 52–54). We note the remarkable property of weakly absorbing dielectric media, where at resonance, for example, Ge resonators can absorb significantly more light than those that intersect their physical cross sections. As a result, light can be absorbed by the spatially isolated (localized) disks when the incident light is spectrally aligned with the optical resonances supported by the nanostructures. Therefore, this results in a modified reflection spectrum and color appearance.

Subwavelength localization of light is naturally provided by plasmonic nanostructures (55), whereas high–refractive index dielectrics may also manipulate light on length scales beyond its wavelength (39, 56, 57). The resonant wavelengths in high-index dielectric cavities intuitively follow the picture of whispering gallery–like modes, where the effective wavelength of the trapped light is related to the geometry of the cavities. Here, we note an important difference with plasmonic structures that share the same overall geometry (45), whereas the dielectric film is replaced by a metallic film. In the plasmonic case, localized disk resonances...
hybridize through near-field coupling with hole plasmon resonances that are also spatially localized (32), that is, with a mutual spatial overlap. On the other hand, in our dielectric case, the holey film acts as a void-type slab photonic crystal (58), with delocalized low-frequency states predominantly occupying the high-index region of the layer while hardly penetrating into the low-index hole regions. Consequently, disk modes have a modest field overlap with modes in the photonic crystal film, and the response can largely be understood from the disk dipole modes supported by the array of dielectric disks. In addition to electric dipole resonances, magnetic dipole resonances can also be supported in these structures (59–61). This property makes the high-index dielectric resonators a diverse system with additional degrees of freedom. For example, by controlling the height and radius of the disk, one may achieve tunable double-resonance characteristics (61), a property that may help further improve the spectral response and, in turn, the color appearance. For the realization of large-area metasurfaces, these dielectric nanoparticles can be printed onto surfaces using laser-based methods (62), whereas we lean against approaches developed for the production of plasmonic color metasurfaces (32, 46).

By systematically varying the disk parameters, such as the disk radius $R$ and the film thickness $T$ (while keeping the height of the pillar $H = 60 \text{ nm}$ fixed), we achieve a full palette of colors that span the visible range, as shown in Fig. 1C. Figure 1D shows the related color gamut of these color palettes in a standard CIE-1931 color space. Although the colors are not at the edges, they still cover most of the color space for traditional laser color printers. The measured reflection spectra for Ge films of thickness $T = 35 \text{ nm}$ coated on the pillars with $R = 30$ to $90 \text{ nm}$ are shown in Fig. 1E (left), demonstrating broadband absorption resonances, with the spectral position determined by the radius of the disks. The wide optical absorption band creates the high-contrast color by suppressing a portion of the reflectivity in the visible spectrum down to the 5% reflection level. The corresponding simulated reflection spectra are shown in Fig. 1E (right), showing qualitative agreement with the experimental results. It should be mentioned here that one particular challenge of plasmonic colors relates to vibrancy, in which all-dielectric nanostructures are potentially supporting resonances that are narrower than plasmonic counterparts, where ohmic loss gives some inherent limitations to the resonance quality factor (63). In this context, Ge represents an alternative, where the homogeneous broadening of resonances is acceptable, whereas there is sufficient above-bandgap absorption for the laser heating to work effectively (fig. S2). Moreover, a protective coating will slightly redshift the resonances in our system, thereby causing a relatively small color change as a result of the increased surrounding refractive index (fig. S3). To appreciate the importance of localized resonances within the subwavelength structures, we preserved the optical properties with respect to the angle of light incidence. Figure 1F shows images of a sample taken at normal and large oblique angle luminescence. Because the resonators are smaller than the wavelength of light, the supported localized resonances ensure that the reflected colors from these structures are robust with respect to the angle of incidence. We further demonstrated this by measuring and simulating the angle-resolved reflectivity of a sample, showing that the...
absorption feature remains prominent for angles of incidence from 
normal incidence up to 60° (see Fig. 1G).

In addition to the structure-generated static colors, there is also 
an interest in postprocessing and dynamical reconfiguration of colors 
(48, 64–66). Plasmon resonances in metal nanostructures offer the 
capability of confining the optical excitation far beyond the diffraction 
limit (55, 67). This can be used to greatly enhance the photothermal 
effect (68), and it was explored for plasmonic color laser printing, where 
plasmonic heating locally melts and reconfigures the metallic nano-
structures (69). Obviously, plasmonic heating is closely linked to field 
confinement at metal surfaces, with limited access to structural melting 
that emerges from a material’s bulk response. Furthermore, plasmonic 
resonances may also provide the particles with a repulsive momentum 
from the surface (70), leading to a random offset of the particles at printing 
(69, 71). In optical heating, the absorbed heat power density reflects 
the electric field distribution $\mathbf{E}$ in the resonator, that is, $q_{\text{abs}} = \frac{1}{2} \epsilon' \omega |\mathbf{E}|^2$, 
where $\omega$ is the angular frequency and $\epsilon'$ is the imaginary part of the 
dielectric function $\epsilon = \epsilon' + i \epsilon''$ of the resonator material. Optically, because 
they are largely transparent at the frequencies of interest, dielectrics 
differentiate themselves from metals. Hence, we usually associate di-
electrics of high optical quality with $\epsilon' > \epsilon'' = 0$. In this aspect, semi-
conductors, such as Si and Ge, are high-index dielectrics with $\epsilon' \gg 1$, 
whereas above-bandgap absorption provides a significant damping in 
the visible region, where $\epsilon''$ for Ge is comparable to that of Al in the 
visible region. Thus, we anticipate opportunities for localized dielectric 
heating and melting-induced morphology changes of Ge nanostruc-
tures, with possible parallels to plasmonic heating used for plasmonic 
color laser printing (69). Here, the more evenly distributed light energy 
in the dielectric resonators will potentially heat the resonator volumes 
more homogeneously. The associated heat-driven morphology changes 
might, in turn, be more controllable, thus allowing us to realize the long-
standing goal of ink-free, high-speed, and subdiffraction full-color laser 
printing in an entirely dielectric material architecture.

Laser heating is generally of a random stochastic nature because of 
the complex thermodynamic phase transitions in the materials (72, 73). 
When the incident laser frequency is close to an optical resonance of the 
structure, the frequency coincidence (or resonance) can promote greatly 
enhanced absorption of electromagnetic energy, which facilitates the 
ultralow power consumption and frequency selectivity in laser printing. 
Consider a morphology-dependent optical resonance, which is reso-
nantly driven by a pulsed laser. Because the resonance frequency de-
deps critically on the detailed morphology of the resonator volume, the 
on-resonance pulsed laser should only provide a little energy dissipation 
to slightly change the morphology, that is, from an initially flat disk in 
the direction of a spherical shape, thus shifting the spectral response into 
an off-resonance state. The heating is consequently weakened, and mor-
phology reshaping is discontinued (fig. S4). This optical resonance-
based selectivity provides distinct power targets for generating identical 
structural colors by well-controlled photothermal reshaping processes, 
which offer easier color selection and halftoning. Figure 2A illustrates

![Fig. 2. Macroscopic RLP of structural colors. (A) Schematic setup of RLP. Synchronous motion solution with the laser pulses is provided by computer-controlled motor or piezo stages. (B) Reflection and transmission microimages of multicolored structures generated by gradually increasing laser powers. Microstructures (i to ix) are generated under gradually increasing laser power strengths from 0.2 to 1.8 μJ in steps of 0.2 μJ, controlled using a liquid crystal attenuator. Scale bars, 0.5 mm. (C) Corresponding SEM images of the microstructures (i, iii, v, vii, and ix) in (B), showing the change of the morphology of the unit cell from disk to sphere and eventually a hole. Scale bars, 200 nm. (D) Experimental reflection spectra of the corresponding laser-printed color squares in (B). Note that the dips (absorbed light) with varying printing laser powers blueshift, causing color changes from cyan to yellow, a.u., arbitrary units. (E) Corresponding simulated reflection spectra. (F) Gamut loop of laser-printed structural colors, which covers CMY colors. (G) Collection of laser-printed paintings with protective polymethyl methacrylate (PMMA) coating (sufficiently thick to suppress Fabry-Pérot interference) deposited before laser printing. Samples have $R = 135$ nm with deposited Ge film thicknesses $T = 45$ nm (i), 40 nm (ii), 35 nm (ii), and 30 nm (iv), respectively. Scale bars, 5 mm. (H) Image of a colorful portrait (i), CMY halftoning of the original image (ii), printing of halftoned magenta color onto the initial cyan background (iii), and laser-printed full-color CMY halftoned image (iv). Scale bars, 5 mm. (I) Structural color laser printing of a portrait image. CMY halftoning realizes red, brown, and especially a saturated black color.](https://advances.sciencemag.org/content/3/5/ea602487/g5)
the setup for resonant laser printing (RLP) of structural colors. We used the selectivity of RLP to achieve resonance-mediated structural color printing governed by photothermal reshaping. Figure 2B presents the reflective and transmitted optical microscope images of the resulting colors, showing gradual color change from cyan to yellow when the applied laser irradiation with single nanosecond laser pulses is linearly increased (pulse energy of 0.2 to 1.8 μJ in the focal plane). We acquired SEM images after laser heating using different energy doses (Fig. 2C).

The laser heating within the Ge resonators gives well-defined reshaping of structures, whereas plasmonic laser heating, as used recently for plasmonic color printing (69), might result in particles that are propelled away from the interface owing to the intense laser field and a rapid thermal change at the metal surface during the melting process (see also fig. S5). Figure 2D and 2E show the measured and simulated spectral data corresponding to each RLP in Fig. 2B (color squares). Spectral features blueshift as a result of the gradual decrease in the disks’ diameter. The related colors generated by RLP can cover cyan, magenta, and yellow (CMY) gamut, as represented in a CIE-1931 chromaticity diagram (Fig. 2F).

To demonstrate the working principle of RLP, we printed several macroscopic images in various color tones, which are shown in Fig. 2G. All images are visible and can be recorded using a commercial camera (Canon EOS) (see also fig. S6). The color appearance and contrast demonstrate the feasibility of our RLP scheme on these ultra-thin, flexible, and laser-printable metasurfaces that work as printable paper (see also fig. S7).

We also demonstrate the full CMY color halftoning abilities of our RLP approach, which is a normal component of traditional laser printers. The full capability of the structural color laser printing is exemplified in Fig. 2H by printing a colorful portrait of Niels Bohr. The original (Fig. 2H, i) and the corresponding CMY halftoning (Fig. 2H, ii) exhibit a mixture of all three base CMY colors. During the RLP process, the intensity of the magenta color is gradually tuned on the cyan background (Fig. 2H, iii). The final full-color printed portrait (Fig. 2H, iv) closely matches the original image (Fig. 2H, i). In most commercial ink-jet or laser printers, a black K (key) ink is also included, resulting in the CMYK color scheme. The K ink is needed for improving the image sharpness due to the imperfect transparency of commercially practical CMY inks. However, structural color printing processes almost never include a black component (which would imply total optical absorption) (45). For structural colors with high-index lossy dielectrics, the intrinsically large optical attenuation within the material leads to already strong resonant absorption (resulting in high-contrast colors) inside the structured films, which can be as thin as tens of nanometers. In our structural color laser printing processes, the printed CMY colors are almost perfectly reflective. A highly saturated CMY combination successfully leads to a halftoned black color, as shown in Fig. 2I. Our results have immediate implications for ink-free color decoration in industrial labeling and packaging, whereas we also anticipate its use for visual commodities, arts, and personal accessories with customized designs.

Super-resolution printing for color filtering has possible applications in retina displays for portable devices. We achieve a subdiffraction
printing resolution of RLP, which is eventually limited by the lattice constant \( P \) of our metasurface. Our particular case with a 200-nm pitch translates into a resolution in excess of 120,000 dots per inch (DPI), with a corresponding pixel that is up to 500-fold smaller than the cross section of a human hair. We now discuss how we address a single resonator in our RLP (see Fig. 3A). The resolution of a traditional microscope used in laser printing (based on a lens for visible light) is limited by diffraction to \( \sim \lambda/2 \) NA, where, in our case, \( \lambda = 532 \) nm, NA is the numerical aperture, and NA \( \leq 0.85 \) (Fig. 3A, i). RLP easily exceeds the diffraction limit of conventional optical microscopy. The localized resonators provide a large field enhancement (Fig. 3A, ii) and a periodical redistribution of the optical energy densities by confining the optical energy within the high-index volumes (with high density of optical states), which are far beyond the diffraction limit. This spatial power enhancement and modulation of the Gaussian-like beam allows an effective printing area with a subdiffraction limited scale because of a threshold in the energy that is needed to drive morphology changes (Fig. 3A, iii). Thus, we can print an ultimate structural color pixel by only addressing the resonator associated with a single unit cell, whereas the energy dissipated in neighboring unit cells would not exceed the threshold for morphology changes (see Fig. 3B and fig. S8). To demonstrate the subdiffraction resolution color laser printing capability, we drew massive patterns on samples with \( T = 35 \) nm and \( R = 70, 80, \) and 90 nm. Figure 3C shows a collection of images printed with a subdiffraction resolution of 127,000 DPI. RLP was carried out at a speed of 1 min per image (only limited by the translation of the mechanical stage; see also movie S1). The images (with 62,500 pixels) are 50 \( \mu \)m by 50 \( \mu \)m each, which are even smaller than a single pixel of the latest Retina HD Display technology (63 \( \mu \)m). With cutting-edge RLP technology, the homogeneous field distribution of resonances inside the particles makes it possible to rapidly heat and modify these nanostructures with incredible accuracy. As shown in the SEM image in Fig. 3D, our RLP of the Parthenon temple discloses the laser-printed subdiffraction features that are beyond the reach of traditional microscopes. Figure 3E shows a high-magnification image (subarea with fine features indicated in Fig. 3D), illustrating the morphology transition of each particle due to the single-unit cell laser printing.

To highlight the potential for future applications, particularly for three-dimensional display or encryption purposes (74), we used a polarization-sensitive color palette to create complex images by RLP at resolutions beyond the diffraction limit. By elongating the disks into bars (Fig. 4A), we show that the asymmetric structures can support tunable color under polarized incident light in the visible spectrum, in a similar manner to plasmonic structures (74, 75). When the local
cylindrical rotational symmetry is lifted, the response of our electric and magnetic dipole modes becomes polarization-dependent, and colors can be switched continuously through polarization tuning (Fig. 4B). The application of RLP to asymmetric structures can easily render asymmetric resonators into morphologies of higher symmetry, that is, bars into spheres, because the latter are favored by a lower surface energy. The transition from Ge bars to spheres and eventual ablation (Fig. 4C) offers new degrees of freedom for structural color applications (fig. S9). Examples of laser-printed patterns with polarization-dependent color appearances are shown in Fig. 4D. These patterns have been printed with a resolution of 100,000 DPI (limited by the unit cell of 250 nm by 250 nm). Polarization-dependent laser printing allows for the encoding or hiding of color information within a single layer of structured surface (movie S2). By varying the Ge film thickness (here, \( T = 30 \) nm), we can modify the polarization-dependent optical responses of the laser-printed structures, making the optical resonances of transverse modes for bars, shortened bars, spheres, and holes located at the same frequencies, resulting in the same color printings. As shown in Fig. 4E, the printed pattern vanishes within the brownish background when the surface is illuminated by 90° polarized light, because all resonances produce colors that resemble the current color of the background (movie S3). We believe that RLP of a full-color high-resolution image, which can markedly change its appearance under orthogonal polarization illuminations, may pave the way for new functional color displays, security labels, and optical data storage applications.

**CONCLUSIONS**

In summary, we have demonstrated a unique structural color platform based on high-index dielectrics. Our realization of non-iridescent structural colors paves the way for a new class of materials for structural color generation, which are ultrathin, fully flexible, and environment-inert. In contrast to plasmonic heating, our work harvests from above-bandgap absorption, which represents the first applications of laser-printable structural colors with all-dielectric materials. With appropriately designed dielectric resonances, sufficient optical attenuation in the visible range allows resonant absorption of laser pulses. Thus, in practice, instantaneous heating and subsequent morphology relaxation of resonators allow ultrafast laser printing of full-color images with ultra-high resolution. We demonstrate the high selectivity of printed colors by RLP, with examples of pure color printing and high-contrast CMY half-tonings. Although we have used the visible absorption of Ge, RLP would also work with absorption bands beyond optical frequencies, whereas the "full" transparency of the visible band itself would facilitate even higher color vibrancy. Furthermore, we demonstrate the ultimate print resolution offered by RLP. Because of the subdiffraction spatial field modulation from the periodically arranged localized resonators, these structures enabled the printing of colors with a resolution exceeding 100,000 DPI, with the possibility for polarization-controlled colors. Our demonstrations open up new potential applications ranging from packaging and displaying to encryption, data storage, and retina devices.

**MATERIALS AND METHODS**

**Sample preparation**

To fabricate the Si master for nanoimprint, a resist-coated Si wafer was exposed by electron-beam lithography (JEOL JBX-9500FS 100 keV prototype) and then dry-etched. The Si stamp coated with an anti-stiction coating was used to replicate the pillar structure via room temperature nanoimprint into the Ormocomp layer. The Ormocomp film was cured by exposure to ultraviolet light, separated from the Si master, and peeled from the Borofloat glass substrate. Subsequently, a Ge film was deposited by an electron beam evaporator at 5 Å/s after imprint under a process pressure of ~10⁻³ mbar. A sufficiently thick layer of PMMA (10% 950PMMA in anisole, MicroChem Corp.) could be spin-coated on top to protect the samples.

**Optical setups**

The optical setup for laser color printing comprised a Nikon Ti-U inverted microscope, where a laser (CryLaS FDSS352-150) emitting 1-ns pulses at 532 nm was used for printing. The laser pulse energy was controlled and rapidly modulated with a liquid crystal attenuator. The sample was mounted on a computer-controlled piezo nanopositioning stage (Mad City Labs Nano-H50 series piezoelectric stage, 0.1-nm resolution, 50-μm travel, XY axis), which was placed on a computer-controlled motor stage. The same optical setup was used for spectroscopic imaging and measurements by combining an imaging spectrometer with a grating of 300 g/mm (Andor Shamrock 303i and Newton 920 CCD with 256 x 1024 pixels). Optical micrographs were acquired using a Nikon Eclipse L200 microscope with a Nikon digital camera (DS-Fi1). Macrophotographs were taken using a commercial digital camera (Canon EOS Mark II series).

**Numerical simulation**

The optical response of our structures was simulated in a commercially available electrodynamic simulation package (CST microwave studio) using a finite element method for reflection spectra and field distributions of structures under different illumination conditions. To simulate the photothermal reshaping processes, we used a simplified model of the complex thermodynamic phase transition. Neglecting gravity, we varied the thickness of round-cornered disks (to the final spherical shape) while preserving the overall initial material volume of the disks. An EM-thermal cosimulation approach was further used to solve the coupled electromagnetic and heat transfer problems.

**SUPPLEMENTARY MATERIALS**

Supplementary material for this article is available at http://advances.sciencemag.org/cgi/content/full/3/5/e1602487/DC1

fig. S1. Permittivities of deposited amorphous Ge and Si and structural colors with silicon resonators.

fig. S2. Ge resonator versus Al resonator.

fig. S3. Radius-dependent resonances and colors for uncoated and coated samples.

fig. S4. Simulations of the heat power generation and transferring within a resonant unit cell.

fig. S5. Comparison of plasmonic laser printing with nonplasmonic RLP.

fig. S6. Reflective and semitransparent colors that resulted from RLP on different samples.

fig. S7. Flexible and robust samples.

fig. S8. A collection of samples with a color printing resolution of 127,000 DPI.


movies S1 to S3

Reference (76)

**REFERENCES AND NOTES**


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